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# Effects of rare earth chlorides on the preparation of porous ruthenium oxide electrodes

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## Abstract

 $RuO_2-RE_2O_3/Ti$  (Ru:RE 7:3) electrodes prepared from the thermal decomposition of RuCl<sub>3</sub> and RECl<sub>3</sub> were much more porous than conventional RuO<sub>2</sub>/Ti electrodes. Porosities of the RuO<sub>2</sub>-RE<sub>2</sub>O<sub>3</sub>/Ti electrodes depended on the ionic radii of the rare earth ions. Rare earth ions added to electrodes to prepare the RuO<sub>2</sub>-RE<sub>2</sub>O<sub>3</sub>/Ti electrodes were completely dissolved in the acid solution. After dissolution of the rare earth ions, the voltammetric charges of the resulting RuO<sub>2</sub>/Ti electrodes were still large. Rare earth chlorides were effective as pore initiators in preparing porous RuO<sub>2</sub>/Ti electrodes, and were recovered after the preparation as catalysts.

Keywords: Ruthenium oxide; Rare earth chloride; Porous material; Pore initiator

# 1. Introduction

Ruthenium oxide electrodes such as RuO<sub>2</sub>/Ti electrodes have been developed for the industrial uses of chlorine generating anodes in the chlorine-alkali process and oxygen generating anodes in electrolytic plating [1]. Ruthenium oxide electrodes have characteristically larger capacitances than metal electrodes. The application of large capacitances of ruthenium oxide electrodes to a supercapacitor system has been proposed [2,3]. The magnitude of capacitances depends on both the surface area and the specific voltammetric charge of a unit surface area of ruthenium oxides. Conventional RuO<sub>2</sub>/Ti electrodes were made up of rather large RuO<sub>2</sub> particles (ca. 10-100 nm) and thereby insufficiently large surface areas. Capacitances are enhanced by dispersing ruthenium oxides with matrix oxides such as TiO<sub>2</sub>, SnO<sub>5</sub> and Ta<sub>5</sub>O<sub>5</sub>. Although the matrix oxides are effective in reducing oxide particle sizes and enlarging surface areas, the specific voltammetric charge of a unit surface area also changes because of differences in electrode composition. Porous electrodes containing single ruthenium oxides are required for the characterization and design of ruthenium oxide electrodes. A novel method should be required for preparing porous RuO<sub>2</sub>/Ti electrodes.

Recently, novel mesoporous carbons have been prepared by activating the carbon pitch with a rare earth ion [4]. Rare earth ions would be effective for preparing porous materials. We have found porous structures of RuO<sub>2</sub>-La<sub>2</sub>O<sub>3</sub>/Ti electrodes prepared from the thermal decomposition of RuCl<sub>3</sub> and LaCl<sub>3</sub> (Ru:La 7:3) [5]. Lanthanum ions of the  $RuO_2-La_2O_3/$ Ti electrodes are completely dissolved in  $H_2SO_4$ ; the porous structure remains unchanged during dissolution. Thus, porous RuO<sub>2</sub>/Ti electrodes can be prepared from the RuO<sub>2</sub>-La<sub>2</sub>O<sub>3</sub>/Ti electrodes. Voltammetric charges of the porous RuO<sub>2</sub>/Ti electrodes were about 20 times larger than those of conventional RuO<sub>2</sub>/Ti electrodes prepared by the thermal decomposition method. Lanthanum chloride plays an important role in preparing porous RuO<sub>2</sub>/Ti electrodes.

In this study, a variety of rare earth chlorides was used to prepare porous  $RuO_2/Ti$  electrodes. The dependences of voltammetric charges and morphology on the ionic radii of rare earth ions were studied.

#### 2. Experimental

The substrates of electrodes used in this study were commercial 99.5% titanium plates,  $10 \times 10 \text{ mm}^2$  in

size, and rods, 1.6 mm in diameter. They were first etched in 10% oxalic acid solution at 80°C for 1 h, then washed with distilled water and dried. The electrodes were dipped into a mixture of n-butanol solution of 0.5 mol dm<sup>-3</sup> RuCl<sub>3</sub> nH<sub>2</sub>O (Ru 45.3%) and ethanol solution of 0.5 mol dm<sup>-3</sup> RECl<sub>3</sub> nH<sub>2</sub>O. After drying the substrates at 60°C for 10 min, they were calcined at 450°C in a preheated furnace for 10 min to decompose the metal salts into their respective oxides. For each specimen, this treatment was repeated 10 times.

The electrochemical cell consisted of a beaker equipped with a working electrode, a platinum plate counter electrode, and a silver-chloride reference electrode. The geometric surface area of the working electrode was  $0.52 \text{ cm}^2$ . A cyclic voltammogram was recorded by sweeping the electrode potential at 50 mV s<sup>-1</sup> in a 0.5 mol dm<sup>-3</sup> H<sub>2</sub>SO<sub>4</sub> aqueous solution. Electrode potentials were referred to the reversible hydrogen electrode (RHE) scale. A Luggin capillary faced the working electrode at a distance of 2 mm. Voltammetric charges ( $q^*$ ) corresponding to active surface areas were determined by integrating the enclosed area of the cyclic curves between 0.30 and 1.10 V (Fig. 1). The surface morphology of the elec-

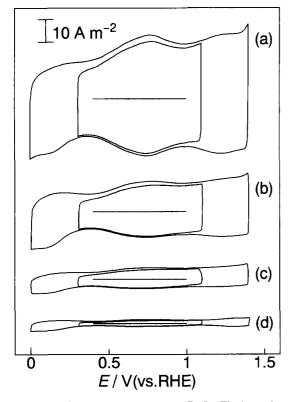


Fig. 1. Cyclic voltammograms for novel  $RuO_2/Ti$  electrodes prepared from (a)  $RuO_2-La_2O_3/Ti$  (Ru:La 7:3), (b)  $RuO_2-Eu_2O_3/Ti$  (Ru:Eu 7:3) and (c)  $RuO_2-Yb_2O_3/Ti$  (Ru:Yb 7:3) electrodes compared with (d) that for a conventional  $RuO_2/Ti$  electrode in 0.5 mol dm<sup>-3</sup>  $H_2SO_4$  swept at 50 mV s<sup>-1</sup>.

trodes was observed using a high resolution scanning electron microscope (Jeol JSM-6000F).

# 3. Results and discussion

Cyclic voltammograms at the steady state in  $H_2SO_4$ for RuO<sub>2</sub>-La<sub>2</sub>O<sub>3</sub>/Ti (Ru:La 7:3), RuO<sub>2</sub>-Eu<sub>2</sub>O<sub>3</sub>/Ti (Ru:Eu 7:3) and RuO<sub>2</sub>-Yb<sub>2</sub>O<sub>3</sub>/Ti (Ru:Yb 7:3) electrodes are compared with those for conventional RuO<sub>2</sub>/Ti electrodes prepared by the thermal decomposition method. Voltammetric charges  $q^*$  for  $RuO_2-La_2O_3/Ti$  (Ru:La 7:3) electrodes were one order of magnitude higher than those for conventional electrodes. Even RuO<sub>2</sub>-Yb<sub>2</sub>O<sub>3</sub>/Ti (Ru:Yb 7:3) electrodes have still larger surface areas than those for conventional electrodes. The order for voltammetric charges of RuO<sub>2</sub>-RE<sub>2</sub>O<sub>3</sub>/Ti (Ru:RE 7:3) electrodes is essentially the same as that obtained for radii of rare earth ions in electrodes. Fig. 2 shows the dependence of voltammetric charges of RuO<sub>2</sub>-RE<sub>2</sub>O<sub>3</sub>/Ti (Ru:RE 7:3) electrodes on the radii of rare earth ions. The voltammetric charge increased with increasing rare earth ion radius.

It was noted that no  $\text{RuO}_2-\text{Re}_2\text{O}_3/\text{Ti}$  (Ru:RE 7:3) electrode contained any rare earth ion after reaching the steady state of cyclic voltammograms. Rare earth ions were completely dissolved in H<sub>2</sub>SO<sub>4</sub> to form novel RuO<sub>2</sub>/Ti electrodes at the steady state of cyclic voltammograms. The voltammetric charge depended on both the surface area and the specific charge on a unit surface area. Specific charges on a unit surface area of both novel and conventional RuO<sub>2</sub>/Ti electrodes would be equivalent because the novel electrodes after dissolution of rare earth ions have the same composition as conventional electrodes. Therefore, novel electrodes have a much larger surface area

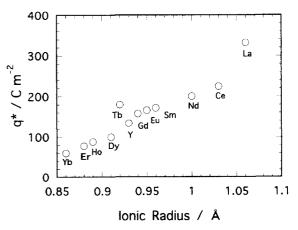


Fig. 2. Dependence of rare earth ion radii in precursor  $RuO_2$ - $RE_2O_3/Ti$  (Ru:RE 7:3) electrodes on voltammetric charges of resulting porous  $RuO_2/Ti$  electrodes in 0.5 mol dm <sup>3</sup> H<sub>2</sub>SO<sub>4</sub> swept at 50 mV s<sup>-1</sup>.

than conventional electrodes. Rare earth ions are effective for preparing porous structures.

Voltammetric charges of  $RuO_2-Re_2O_3/Ti$  (Ru:Re 7:3) electrodes were also large before the dissolution of rare earth ions. Porous structures are formed not by the dissolution of rare earth ions, but during the thermal decomposition with rare earth chloride. Fig. 3 shows microstructures of  $RuO_2-La_2O_3/Ti$  (Ru:La 7:3),  $RuO_2-Eu_2O_3/Ti$  (Ru:Eu 7:3) and  $RuO_2-Yb_2O_3/Ti$  (Ru:Yb 7:3) electrodes before the dissolution of rare earth ions. The most porous structure was observed in  $RuO_2-La_2O_3/Ti$  (Ru:La 7:3) electrode, which exhibited the maximum voltammetric charge (Fig. 3(a)). The order of porosity was the same as that obtained

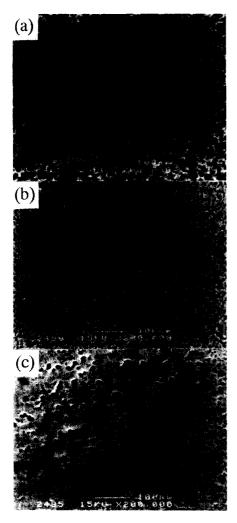


Fig. 3. High-resolution scanning electron micrographs of (a)  $RuO_2-La_2O_3/Ti$  (Ru:La 7:3), (b)  $RuO_2-Eu_2O_3/Ti$  (Ru:Eu 7:3) and (c)  $RuO_3-Yb O_3/Ti$  (Ru:Yb 7:3) electrodes.

for voltammetric charge. Large rare earth ions such as a lanthanum ions enlarged the surface area. The size of rare earth ions determined the pore size of the  $RuO_2/Ti$  electrodes. It is likely that rare earth chlorides act as pore initiators. The pore sizes evaluated from Fig. 3 were larger than the ion sizes. The pore formation is caused not by the dissolution of rare earth ions, but by disordering ruthenium oxide clusters during thermal decomposition with rare earth chloride. The coordination of rare earth ions to ruthenium oxide clusters is critical for the pore formation. The mechanism of pore initiation should be further studied.

The chemical properties of rare earth ions are similar to those of alkali earth ions. Voltammetric charges for RuO<sub>2</sub>-CaO/Ti (Ru:Ca 7:3) electrodes are rather larger than RuO<sub>2</sub>-La<sub>2</sub>O<sub>3</sub>/Ti (Ru:La 7:3). SEM images show, however, that RuO<sub>2</sub>-CaO/Ti electrodes have a much less porous structure than  $RuO_2-La_2O_3/$ Ti (Ru:La 7:3). Calcium ions are not effective for pore formation, although they enlarge the specific charge on a unit surface area. Moreover, calcium ions in RuO<sub>2</sub>-CaO/Ti (Ru:Ca 7:3) electrodes were not dissolved in any acid solution. Calcium ions are incorporated into the bulk structure of oxide electrode. However, rare earth ions in oxide electrodes can be recovered from acid solutions which dissolve rare earth ions. Rare earth ions will act as catalysts for preparing porous oxide materials.

#### Acknowledgements

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